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Effect of dielectric discontinuity on a spherical polyelectrolyte brush

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In this paper we perform molecular dynamics simulations of a spherical polyelectrolyte brush and counterions in a salt-free medium. The dielectric discontinuity on the grafted nanoparticle surface is taken into account by the method of image charges. Properties of the polyelectrolyte brush are obtained for different parameters, including valency of the counterions, radius of the nanoparticle, and the brush total charge. The monovalent counterions density profiles are obtained and compared with a simple mean-field theoretical approach. The theory allows us to obtain osmotic properties of the system. *Published by AIP Publishing.* [<http://dx.doi.org/10.1063/1.5002526>]

I. INTRODUCTION

The study of polyelectrolyte chains grafted to surfaces in a structure known as polyelectrolyte brush (PEB) has acquired substantial interest recently, as covered by many reviews.^{1–7} With the development of experiments with DNA molecules outside of the intracellular environment, the study of cell-free gene expression has brought a new horizon for biotechnology. Examples go from double-stranded DNA brushes⁸ to a single-step photolithographic biocompatible DNA mono-layer,⁹ both on a biochip. In addition to these, it is also valid to refer to other applications for the synthesization of PEBs such as protein absorption,¹⁰ bioseparation,¹¹ and targeted drug/gene delivery.¹² When referring to the term brush we assume that the grafting of the chains is dense enough in a way that the linear dimensions of the polyelectrolyte chains are much larger than the average distance between two neighboring charged polymers on the surface.³ Previous studies^{13–16} have shown that an essential property of a PEB is in its capability to confine a major quantity of counterions in a way to compensate its electrical charge, resulting in high osmotic pressure governing its stretching dynamics. Besides their extensive range of applications, PEBs have been studied in a range of different configurations as well, they can be generated either by grafting polyelectrolyte chains to planar^{17–21} or to strongly curved systems as, for example, cylinders or spheres,^{22–27} the last being the focus of our present work. Recently, different types of neutral polymer brushes with interesting properties composed by dipolar ions called zwitterions^{28,29} in the form of polyzwitterions, have attracted attention and deserve citation.

In relation to spherical PEB, we can elicit their main structure as being formed by an inorganic core nanoparticle and an organic layer/shell in the form of polyelectrolyte chains grafted to its surface. As a result of their mechanical stability, high surface area, and ease of synthesis, silica/polymer hybrid nanoparticles have been studied more extensively.^{11,30}

Spherical PEBs also carry a number of advantages in comparison with planar ones. They can be studied by a wide variety of distinct methods coined for colloidal particles investigation, from scattering methods^{31–34} to, more recently, dielectric spectroscopy.³⁵ Furthermore, the colloidal dimensions of the spherical PEB may be used to create well-defined surfaces of the order of many m^2 that can be used for nanoparticle/protein immobilization,^{36,37} and they can also be viewed as models for the study of carboxylated latex particles that constitute a major industrial product.³⁸

Amongst previous works, we can cite efforts to theoretically describe PEBs.^{39–41} Regarding molecular dynamics (MD) simulations of spherical PEBs we can cite, as few examples, studies on the dependence of the brush thickness due to different parameters and conformations,^{42–44} studies on brush size as a function of chain lengths, salt concentrations⁴⁵ and grafting densities—these accompanied by comparisons to mean field or self-consistent field theories.^{46,47} The effect of multivalent ions on brush conformations was also extensively studied.^{15,48–50} Nevertheless, MD simulations of PEBs that take into account the dielectric discontinuity between the grafted nanoparticle and surrounding medium are unprecedented so far, to the best of our knowledge. In spite of the preceding statement, the problem of charged particles in heterogeneous dielectric media has been broadly studied resulting in the coinage of different methods. Among those, we refer to treatments that can be applied in the spherical geometry for applications in colloidal science. Even if the computational cost is high, one can use Legendre polynomials^{51,52} to perform MC or MD simulations. A variational formulation has gained attention lately as a more general method for the solution of the Poisson equation treating the local polarization charge density as a dynamic variable.^{53,54} A more efficient method considers the images and uniformly distributed counter-image charges inside the dielectric void as an approximation that works very well for low dielectric constants.^{55,56} In this work, we intend to include nanoparticle polarization using the previously mentioned method and perform MD simulations of a PEB in a salt-free suspension. In addition, a simple Poisson-Boltzmann

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(PB) theory is presented in order to account for the counterions concentration in mean-field regime.

In Sec. II, we explain the model and simulation method followed by the presentation of the theory developed for weak electrostatic coupling. The results are presented in the subsequent section. In the last section we finish describing the conclusions of the present work and general perspectives.

II. MODEL AND SIMULATION METHOD

We follow a standard coarse-grained model for the polymer chains and counterions confined in a spherical cell of radius R . The $N_p = 14$ chains are represented by N_m charged hard spheres (monomers) of radii $r_m = 2 \text{ \AA}$ and charge $+q$, where q is the proton charge. The first monomers of the chains are grafted to the surface of a sphere of radius a and relative dielectric constant ϵ_c , representing the nanoparticle base particle. The first monomers are all uniformly distributed on the nanoparticle surface, grafted at distance r_m from it. The N_c counterions are modeled as hard spheres with effective radii $r_c = r_m$ and charge $-\alpha q$, where α is the valency. The number of counterions is defined as $N_c = N_p N_m / \alpha$ in order to keep the system with zero total charge. The medium in which the polyelectrolyte is immersed is represented by structureless water with relative dielectric constant $\epsilon_w = 80$. The Bjerrum length, defined as $\lambda_B = q^2 / \epsilon_w k_B T$, is 7.2 \AA , the value for water at room temperature.

Following a method previously developed,⁵⁵ we investigate the influence of the nanoparticle polarization by means of image charges. The calculation of image charges for the spherical geometry is not as straightforward as for the planar geometry. The continuity of the tangential component of the electric field and of the normal component of the displacement field across the nanoparticle-water interface, requirements of the Maxwell equations boundary conditions, give rise to a counter-image line charge in addition to the punctual image charge that is the usual requirement for planar geometry.⁵⁷ The electrostatic potential at an arbitrary position \mathbf{r} produced by the arbitrary charge q_i located at \mathbf{r}_i outside the nanoparticle is approximated by

$$\phi(\mathbf{r}; \mathbf{r}_i) = \frac{q_i}{\epsilon_w |\mathbf{r} - \mathbf{r}_i|} + \frac{\gamma q_i a}{\epsilon_w r_i |\mathbf{r} - \frac{a^2}{r_i^2} \mathbf{r}_i|} + \frac{\gamma q_i}{\epsilon_w a} \log \left(\frac{r r_i - \mathbf{r} \cdot \mathbf{r}_i}{a^2 - \mathbf{r} \cdot \mathbf{r}_i + \sqrt{a^4 - 2a^2(\mathbf{r} \cdot \mathbf{r}_i) + r^2 r_i^2}} \right), \quad (1)$$

where $r_i = |\mathbf{r}_i|$, $r = |\mathbf{r}|$, and $\gamma = (\epsilon_w - \epsilon_c) / (\epsilon_w + \epsilon_c)$. This expression is valid⁵⁵ for $\epsilon_w \gg \epsilon_c$. The total electrostatic energy is

$$U_{elec} = \sum_{i=1}^{N-1} \sum_{j=i+1}^N q_i q_j \phi(\mathbf{r}_j; \mathbf{r}_i) + \sum_{i=1}^N \frac{\gamma q_i^2 a}{2\epsilon_w (r_i^2 - a^2)} + \sum_{i=1}^N \frac{\gamma q_i^2 \log(1 - \frac{a^2}{r_i^2})}{2\epsilon_w a}. \quad (2)$$

The two last terms above are the ionic electrostatic self-energy.

The elastic bonds between adjacent monomers of the same chain in the brush are modeled by the following nonlinear

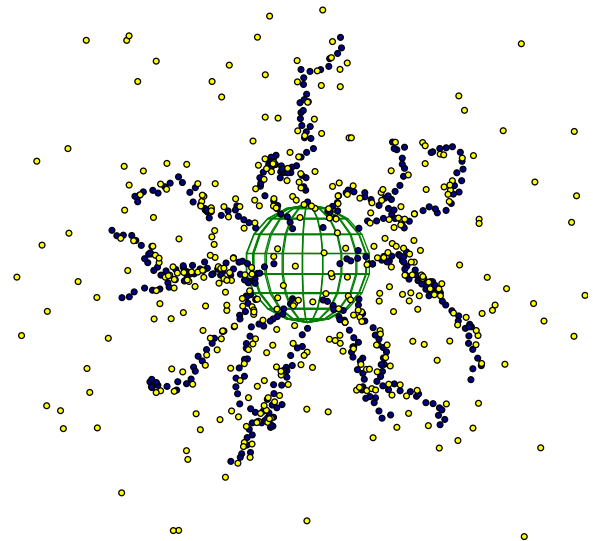


FIG. 1. Representation of the spherical PEB. Darker spheres represent monomers, while lighter spheres represent counterions.

energy potential,^{58–61}

$$U_{bond} = \sum_{ad.mon.} \frac{A}{2} (r - r_0)^2, \quad (3)$$

where $r = |\mathbf{r}_i - \mathbf{r}_j|$ is the distance between adjacent monomers i and j . The sum is made over all adjacent monomers of the same polymer chains, $A = 0.9 k_B T$ and $r_0 = 5 \text{ \AA}$, following the aforementioned Ref. 61.

The total force acting on the charged specie k is

$$\mathbf{F}_k = -\nabla_{\mathbf{r}_k} (U_{elec} + U_{bond}). \quad (4)$$

The molecular dynamics simulations were performed for constant time steps by means of a well-known Langevin equation,⁶²

$$\mathbf{p}'_i(t) = \mathbf{F}_i(t) - \Gamma \mathbf{p}_i(t) + \mathbf{R}_i(t), \quad (5)$$

where $\mathbf{p}_i(t)$ is the momentum of particle i at time t , $\mathbf{F}_i(t)$ is the force felt by this particle, Γ is the friction coefficient, and $\mathbf{R}_i(t)$ is the stochastic force acting on particle i , which satisfy the fluctuation dissipation relation. The Verlet-like method developed by Ermak⁶² is used to solve the previous equation.

The mechanism chosen to avoid the superposition between all particles and surfaces is a hard sphere potential. This was preferred over a capped Lennard-Jones type potential for the reason that the latter was tested showing little to no difference from the hard spheres potential while requiring time steps much smaller to avert simulation crashes. In Fig. 1, a snapshot of MD simulations after equilibrium is shown for monovalent counterions.

III. THEORY

At room temperature, electrostatic correlations between monovalent ions can be neglected.⁶³ A mean-field PB equation is used to obtain the density profile of counterions. We do not consider the dielectric discontinuity on the nanoparticle-water interface because this effect is very small in this regime, as it will be shown in the Results section. Also, the qualitative

results obtained with the present method allow us to consider this approximation. However, it is important to mention that this effect can be important when more accuracy is necessary in the study of the electric double layer.^{55,56}

The charge distribution of PEB is constructed as if all the monomers are aligned with the nanoparticle center, with effective distance between them equals to $r_{ef} = 0.75r_m$ to account for the bending of the chains. The modified PB equation takes the form

$$\nabla^2 \phi(r) = -\frac{4\pi}{\epsilon_w} \left[\sum_{i=1}^{N_m} \sigma_i \delta(r - r_i) - q\alpha \rho(r) \right], \quad (6)$$

where $\phi(r)$ is the mean electrostatic potential, σ_i are the charge densities of the corresponding layers of monomers, given by $\sigma_i = N_p/4\pi r_i^2$, where $r_i = a + r_m + (i-1)2r_{ef}$. The counterions density profile is given by

$$\rho(r) = N_c \frac{e^{-\beta\alpha q\phi(r)}}{4\pi \int_{(a+r_c)}^R dr' r'^2 e^{-\beta\alpha q\phi(r')}}. \quad (7)$$

The solution of Eq. (6) is performed by Picard iterative process.

IV. RESULTS

The results are presented in the form of average particles concentration profiles and average effective PEB radius (the distance between the center of the grafted nanoparticle and the more distant monomer), see Fig. 2. We start by studying the effect of the dielectric discontinuity on the counterion distribution around the PEB, see Fig. 3. We choose the following values for the nanoparticle relative dielectric constant, $\epsilon_c = 2$ and $\epsilon_c = \epsilon_w$. Whereas in the first choice we choose the typical dielectric constant value of silica, in the second case, we ignore the dielectric discontinuity by having the nanoparticle represented by the same material as the medium in which it is inserted, water. Silica nanospheres coated with polymer brushes have already been used for effective separation of glycoproteins.¹¹

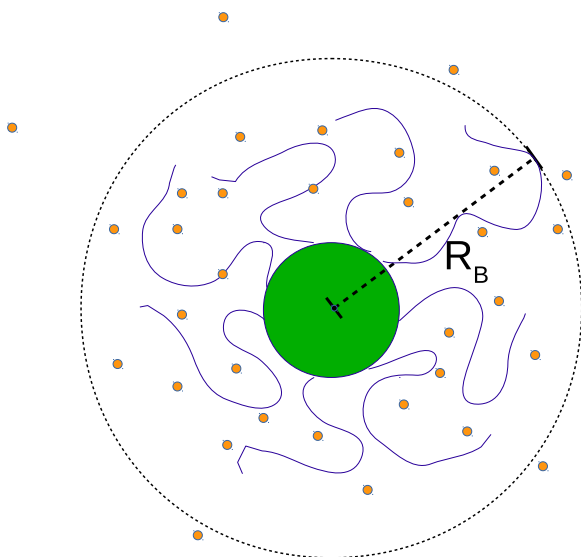


FIG. 2. Definition of effective PEB radius, R_B .

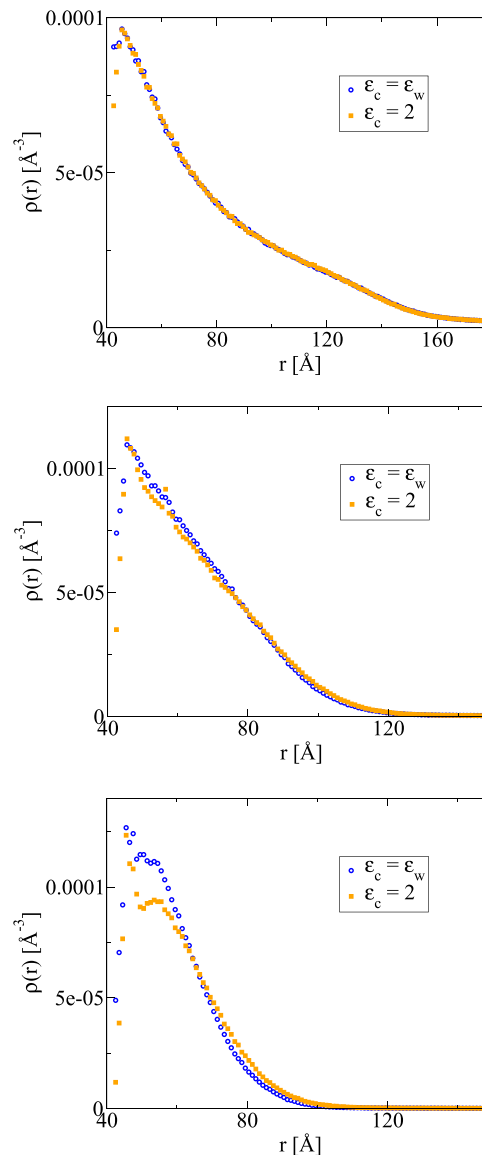


FIG. 3. Density profiles of counterions obtained for $\alpha = 1, 2, 3$, from top to bottom, respectively. Polyelectrolyte brush individual chains with $N_m = 30$ and nanoparticle radius $a = 40$ Å.

The influence of the dielectric discontinuity on monovalent ions is very small and most of the pattern we see is caused by osmotic pressure inside the brush, which tends to repel counterions. Similar brush configuration has been extensively explored before without consideration for dielectric discontinuity so that the results showing that multivalent counterions are more deeply absorbed are expected. Although the density maximum concur for both distributions of multivalent ions, we find that the polarization of the silica nanoparticle tends to broaden their distributions since they feel more repelled by their image charges. Also, the effective brush radius, R_B , tends to be higher due to image charges of chains, which can affect the ionic distribution far away from brush. For charged nanoparticles and surfaces, the consequence of a dielectric discontinuity in ionic distribution is very local, near surfaces.^{54–56,64} The importance of the polarization effect for the trivalent case can also be observed in the density profiles of monomers, see Fig. 4.

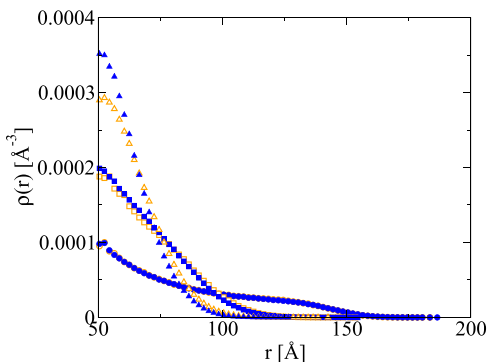


FIG. 4. Density profiles of monomers obtained for $\alpha = 1, 2, 3$, circles, squares, and triangles, respectively. Full symbols represent $\epsilon_c = \epsilon_w$, while open symbols $\epsilon_c = 2$. The parameters are the same as in Fig. 3.

Moving further, we study the brush behavior over different number of monomers and different counterion valence by calculating R_B , see Fig. 5. We define the PEB radius, R_B , as the average distance between the center of grafted nanoparticle and the more distant monomer. Here we confirm that image charges have little to no influence over the brush diameter for monovalent counterions. This is not the case for larger brushes composed by 30 and, more explicitly, 42 monomers, surrounded by multivalent ions. In this case we can find a considerable increase in R_B when accounting for the dielectric discontinuity when compared with the homogeneous case. The polyelectrolyte chain total charge is high for a sufficient number of monomers and they are, by construction, near the nanoparticle surface. This means that image charges play an important role in the brush radius value when this value is sufficiently small. The difference in both approximations can achieve $\approx 9\%$ for the discussed parameters. The smaller values obtained for R_B in the case of multivalent ions are in agreement with experiments that relate the collapse of the spherical PEB with the addition of multivalent ions in solution.^{15,65}

The density profile for a special case in which the brush's nanoparticle is relatively big ($a = 100 \text{ \AA}$) is shown in Fig. 6. The polarization of the nanoparticle undoubtedly has a strong influence over the trivalent counterions profile, showing that the role played by nanoparticle polarization is not only to further fend the colloid and the counterions but also to spread

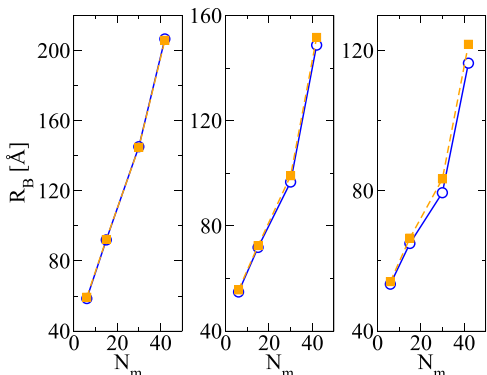


FIG. 5. PEB average effective radius as a function of N_m for $a = 40 \text{ \AA}$ and $\alpha = 1, 2, 3$, from left to right panels, respectively. The circles represent the case, which $\epsilon_c = \epsilon_w$, while squares, $\epsilon_c = 2$.

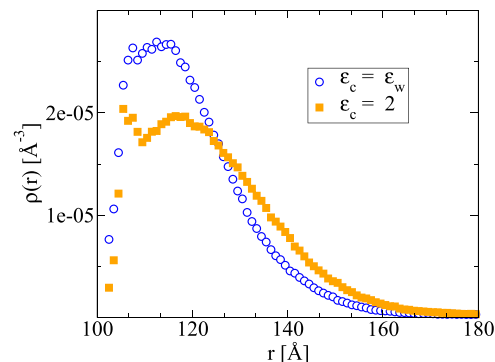


FIG. 6. Density profiles of counterions obtained for $\alpha = 3$ for two relative dielectric constants of the nanoparticle. Polyelectrolyte brush individual chains with $N_m = 30$ and grafted nanoparticle radius $a = 100 \text{ \AA}$.

their distribution, in comparison with the unpolarized nanoparticle. It is also worthy to remark the double peak pattern present in the $\epsilon_c = 2$ curve, much more protruding than in the $\epsilon_c = \epsilon_w$ curve, indicating two clear preferred regions for the trivalent counterions. This is a competition between the electrostatic interaction of multivalent ions with the entire brush and with their local chain, see also Fig. 3, bottom panel. The polarization of the nanoparticle separates more explicitly these regions as a result of the shifting of the ionic distribution.

In order to measure the effect of nanoparticle polarization on counterions distributions as a function of nanoparticle curvature, we calculate the relative difference between profiles

defined as $\Delta = \frac{\sqrt{\int_a^R dr [\rho_2(r) - \rho_{\epsilon_w}(r)]^2}}{\int_a^R dr \rho_{\epsilon_w}(r)}$, where $\rho_{\epsilon_w}(r)$ is the

counterion profile for $\epsilon_c = \epsilon_w$ and $\rho_2(r)$ for $\epsilon_c = 2$. We take the cases of Fig. 3 and similar ones except for the parameters $a = 80 \text{ \AA}$ and $R = 500 \text{ \AA}$ for comparison. We set these lengths in order to maintain constant volume fraction in the comparison. The volume fraction is defined as $\phi_{frac} = a^3/R^3$. For $\alpha = 1$, we obtain for Δ the values 0.0041 and 0.0032. For divalent $\alpha = 2$ sets, we obtain 0.0115 and 0.0166. The values found for $\alpha = 3$ were 0.0265 and 0.0252, all numbers for $a = 40 \text{ \AA}$ and $a = 80 \text{ \AA}$, respectively. These results show us that there is no influence of the nanoparticle curvature in the polarization effect on the counterions distribution for constant PEBs volume fraction. However, if we take for comparison two sets with the same cell radius R but with different nanoparticles radius a , the curvature can decrease the effect of dielectric discontinuity on the counterions distribution. We take the trivalent case of Fig. 3 and the set of Fig. 6. The parameters are the same with the exception of the nanoparticle radius, which is 40 \AA and 100 \AA , respectively. We then obtain the values 0.0265 and 0.0424, respectively, for the relative difference of the profiles, showing that the decrease in curvature enhances the aforementioned effect.

We move further in the results section by comparing simulations with the present theory for monovalent ions, see Fig. 7. The theory is not able to describe properly the monovalent counterions structure around the brush, except for shorter chains. However, the agreement is very good in the region far from nanoparticle surface, for the studied chain lengths. The

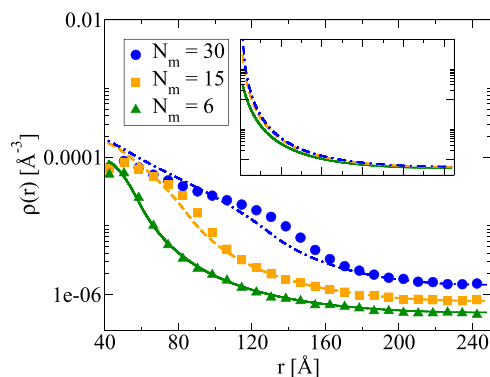


FIG. 7. Density profiles of counterions obtained for $\alpha = 1$, $a = 40$ Å and various number of monomers, N_m . The lines represent the results of the present theory, while symbols represent the results of simulations. The inset shows the solutions of PB equation if all the charged monomers are located on the nanoparticle surface, for the same parameters and the same x and y axis scales.

present method allows us to quantitatively account the adsorption of monovalent counterions, which means that osmotic properties of a brush suspension can be studied using the present method. We can define, for example, effective charges of PEBs, as the subject for a future work. It is important to mention the interesting effect that the boundary ionic concentrations are not saturated with the increase in the macroparticle charge as it is observed in colloidal suspensions; see the inset of Fig. 7. This saturation observed in colloidal suspensions reflects the independence of the colloidal effective charge with the colloidal charge.^{66,67} This is not the case for PEBs as can be seen in Fig. 7. For multivalent counterions, as expected, the theory is not able to describe the asymptotic curve, as can be seen in Fig. 8, not even by reasonably decreasing the value of r_{ef} . The counterion-counterion and counterion-monomer electrostatic correlations take place and the present mean-field theory is not able to account for these effects.

As an application of the method for monovalent counterions, we calculate the osmotic coefficient, which is defined as the fraction between the pressure and ideal pressure given by $\phi_{osm} = p_{bulk}/p_{id}$, where ρ_{bulk} is the counterion bulk concentration and $\rho_{id} = N_c/V$, where V is the volume accessible to the N_c counterions.⁶⁸ In Fig. 9, we show the curves of ϕ_{osm} versus ϕ_{frac} for the same parameters as in Fig. 7 obtained with the

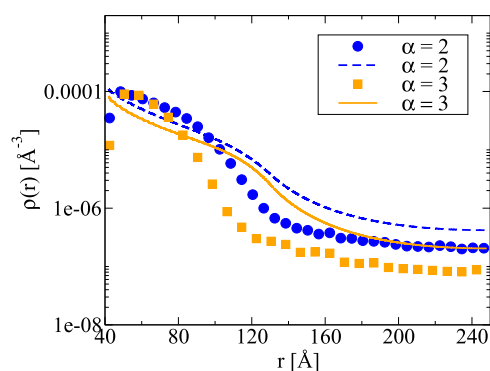


FIG. 8. Density profiles of multivalent counterions obtained for $N_m = 30$ and $a = 40$ Å. The lines represent the results of the present theory, while symbols represent the results of simulations.

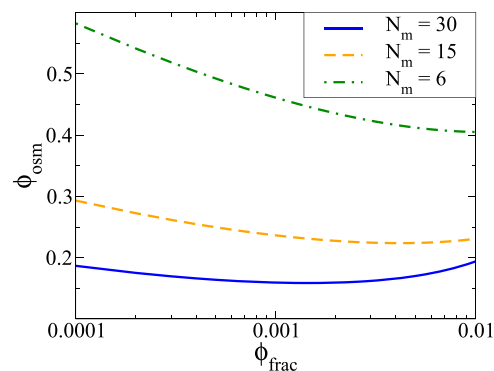


FIG. 9. Osmotic coefficient versus volume fraction for the same parameters of Fig. 7.

present theory. For longer chains, we can observe a minimum in the curve. Also, by increasing the length of grafted chains we obtain a smaller osmotic coefficient that is in agreement with experimental measurements.⁶⁸

V. CONCLUSIONS

In this work we have performed MD simulations of a spherical polyelectrolyte brush in a salt-free solution. The dielectric discontinuity in the grafted nanoparticle surface is taken into account. We observe that for monovalent counterions at room temperature, the grafted nanoparticle polarization is not mandatory to describe the ionic structure around the brush. Also, the effective polyelectrolyte brush radius is not very affected for the studied parameters apart from the cases with trivalent counterions and longer chains, in which, the differences can achieve $\approx 9\%$. Furthermore, in these cases, the concentration profiles of counterions and monomers are considerably different comparing both approximations. We also present a mean-field Poisson-Boltzmann theory for low electrostatic coupling regime. This method allows us to obtain quantitatively the asymptotic counterionic concentration, leading us to calculate the osmotic coefficients of PEBs suspensions. The effective charges of brushes are going to be studied in a future work. We have observed that the nanoparticle curvature influences the polarization effect for constant cell radii. In this comparison the volume available to counterions is in practice the same. However, if we keep the PEB's volume fraction constant in the comparison, the effect of the curvature is negligible. In this comparison the volume available to counterions is very different. This suggests that the effect of the polarization depends also on the ionic strength of the solution.

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¹S. Minko, *Polym. Rev.* **46**, 397 (2006).

²M. Ballauff and O. Borisov, *Curr. Opin. Colloid Interface Sci.* **11**, 316 (2006).

³M. Ballauff, *Prog. Polym. Sci.* **32**, 1135 (2007).

⁴P. Jain, G. L. Baker, and M. L. Bruening, *Annu. Rev. Anal. Chem.* **2**, 387 (2009).

⁵K. Binder and A. Milchev, *J. Polym. Sci., Part B: Polym. Phys.* **50**, 1515 (2012).

- ⁶I. Szilagy, G. Trefalt, A. Tiraferri, P. Maroni, and M. Borkovec, *Soft Matter* **10**, 2479 (2014).
- ⁷S. Das, M. Banik, G. Chen, S. Sinha, and R. Mukherjee, *Soft Matter* **11**, 8550 (2015).
- ⁸E. Karzbrun, A. M. Tayar, V. Noireaux, and R. H. Bar-Ziv, *Science* **345**, 829 (2014).
- ⁹A. Buxboim, S. S. Daube, and R. Bar-Ziv, *Mol. Syst. Biol.* **4**, 181 (2008).
- ¹⁰S. J. Sofia, V. Premnath, and E. W. Merrill, *Macromolecules* **31**, 5059 (1998).
- ¹¹L. D. Jiang, H. Bagan, T. Kamra, T. C. Zhou, and L. Ye, *J. Mater. Chem. B* **4**, 3247 (2016).
- ¹²A. Wittemann, T. Azzam, and A. Eisenberg, *Langmuir* **23**, 2224 (2007).
- ¹³P. Pincus, *Macromolecules* **24**, 2912 (1991).
- ¹⁴O. V. Borisov, T. M. Birshtein, and E. B. Zhulina, *J. Phys. II* **1**, 521 (1991).
- ¹⁵Y. Mei, K. Lauterbach, M. Hoffmann, O. V. Borisov, M. Ballauff, and A. Jusufi, *Phys. Rev. Lett.* **97**, 158301 (2006).
- ¹⁶Y. Mei, M. Hoffmann, M. Ballauff, and A. Jusufi, *Phys. Rev. E* **77**, 031805 (2008).
- ¹⁷H. Ahrens, S. Forster, and C. A. Helm, *Macromolecules* **30**, 8447 (1997).
- ¹⁸M. Biesalski and J. Ruhe, *Macromolecules* **32**, 2309 (1999).
- ¹⁹E. P. K. Currie, A. B. Sieval, M. Avena, H. Zuilhof, E. J. R. Sudholter, and M. A. C. Stuart, *Langmuir* **15**, 7116 (1999).
- ²⁰J. Ruhe, M. Ballauff, M. Biesalski, P. Dziezok, F. Grohn, D. Johannsmann, N. Houbenov, N. Hugenberg, R. Konradi, S. Minko, M. Motornov, R. R. Netz, M. Schmidt, C. Seidel, M. Stamm, T. Stephan, D. Usov, and H. Zhang, *Adv. Polym. Sci.* **165**, 79 (2004).
- ²¹D. Bendejacq, V. Ponsinet, and M. Joanicot, *Eur. Phys. J. E* **13**, 3 (2004).
- ²²L. F. Zhang, K. Yu, and A. Eisenberg, *Science* **272**, 1777 (1996).
- ²³C. Biver, R. Hariharan, J. Mays, and W. B. Russel, *Macromolecules* **30**, 1787 (1997).
- ²⁴P. Guenoun, F. Muller, M. Delsanti, L. Auvray, Y. J. Chen, J. W. Mays, and M. Tirrell, *Phys. Rev. Lett.* **81**, 3872 (1998).
- ²⁵R. Hariharan, C. Biver, J. Mays, and W. B. Russel, *Macromolecules* **31**, 7506 (1998).
- ²⁶X. Guo, A. Weiss, and M. Ballauff, *Macromolecules* **32**, 6043 (1999).
- ²⁷F. Muller, P. Fontaine, M. Delsanti, L. Belloni, J. Yang, Y. J. Chen, J. W. Mays, P. Lesieur, M. Tirrell, and P. Guenoun, *Eur. Phys. J. E* **6**, 109 (2001).
- ²⁸M. Chen, W. H. Briscoe, S. P. Armes, H. Cohen, and J. Klein, *Eur. Polym. J.* **47**, 511 (2011).
- ²⁹M. Kobayashi, Y. Terayama, M. Kikuchi, and A. Takahara, *Soft Matter* **9**, 5138 (2013).
- ³⁰I. Ab Rahman and V. Padavettan, *J. Nanomater.* **2012**, 132424 (2012).
- ³¹W. Groenewegen, A. Lapp, S. U. Egelhaaf, and J. R. C. van der Maarel, *Macromolecules* **33**, 4080 (2000).
- ³²X. Guo and M. Ballauff, *Phys. Rev. E* **64**, 051406 (2001).
- ³³N. Dingenouts, M. Patel, S. Rosenfeldt, D. Pontoni, T. Narayanan, and M. Ballauff, *Macromolecules* **37**, 8152 (2004).
- ³⁴E. Schneck, E. Papp-Szabo, B. E. Quinn, O. V. Konovalov, T. J. Beveridge, D. A. Pink, and M. Tanaka, *J. R. Soc., Interface* **6**, S671 (2009).
- ³⁵X. X. Guo and K. S. Zhao, *J. Phys.: Condens. Matter* **29**, 055102 (2017).
- ³⁶A. Wittemann, B. Haupt, and M. Ballauff, *Phys. Chem. Chem. Phys.* **5**, 1671 (2003).
- ³⁷G. Sharma and M. Ballauff, *Macromol. Rapid Commun.* **25**, 547 (2004).
- ³⁸D. Distler, *Waessrige Polymerdispersionen* (Wiley-VCH, New York, 1999).
- ³⁹Y. Rakita, D. Golodnitsky, and A. Natan, *J. Electrochem. Soc.* **161**, E3049 (2014).
- ⁴⁰A. Erbas and M. O. de la Cruz, *Macromolecules* **49**, 9026 (2016).
- ⁴¹A. Naji, R. R. Netz, and C. Seidel, *Eur. Phys. J. E* **12**, 223 (2003).
- ⁴²F. S. Csajka and C. Seidel, *Macromolecules* **33**, 2728 (2000).
- ⁴³C. Seidel, *Macromolecules* **36**, 2536 (2003).
- ⁴⁴D. J. Sandberg, J. M. Y. Carrillo, and A. V. Dobrynin, *Langmuir* **23**, 12716 (2007).
- ⁴⁵N. A. Kumar and C. Seidel, *Macromolecules* **38**, 9341 (2005).
- ⁴⁶S. Z. He, H. Merlitz, L. Chen, J. U. Sommer, and C. X. Wu, *Macromolecules* **43**, 7845 (2010).
- ⁴⁷H. Merlitz, C. W. Li, C. X. Wu, and J. U. Sommer, *Soft Matter* **11**, 5688 (2015).
- ⁴⁸W. D. Tian and Y. Q. Ma, *J. Phys. Chem. B* **113**, 13161 (2009).
- ⁴⁹L. T. Yan, Y. Y. Xu, M. Ballauff, and A. H. E. M. A. Boker, *J. Phys. Chem. B* **113**, 5104 (2009).
- ⁵⁰L. Liu, P. A. Pincus, and C. Hyeon, *Macromolecules* **50**, 1579 (2017).
- ⁵¹R. Messina, *J. Chem. Phys.* **117**, 11062 (2002).
- ⁵²G. I. Guerrero-García and M. O. de la Cruz, *J. Phys. Chem. B* **118**, 8854 (2014).
- ⁵³V. Jadhao, F. J. Solis, and M. O. de la Cruz, *Phys. Rev. Lett.* **109**, 223905 (2012).
- ⁵⁴V. Jadhao, F. J. Solis, and M. O. de la Cruz, *J. Chem. Phys.* **138**, 054119 (2013).
- ⁵⁵A. P. dos Santos, A. Bakhshandeh, and Y. Levin, *J. Chem. Phys.* **135**, 044124 (2011).
- ⁵⁶A. Bakhshandeh, A. P. dos Santos, and Y. Levin, *Phys. Rev. Lett.* **107**, 107801 (2011).
- ⁵⁷W. T. Norris, *IEE Proc.: Sci., Meas. Technol.* **142**, 142 (1995).
- ⁵⁸R. S. Dias, A. A. C. C. Pais, P. Linse, M. G. Miguel, and B. Lindman, *J. Phys. Chem. B* **109**, 11781 (2005).
- ⁵⁹M. Quesada-Perez and A. Martín-Molina, *Soft Matter* **9**, 7086 (2013).
- ⁶⁰G. Luque-Caballero, A. Martín-Molina, and M. Quesada-Perez, *J. Chem. Phys.* **140**, 174701 (2014).
- ⁶¹A. P. dos Santos, M. Giroto, and Y. Levin, *J. Phys. Chem. B* **120**, 10387 (2016).
- ⁶²M. P. Allen and D. J. Tildesley, *Computer Simulations of Liquids* (Oxford University Press, Oxford, New York, 1987).
- ⁶³Y. Levin, *Rep. Prog. Phys.* **65**, 1577 (2002).
- ⁶⁴A. P. dos Santos and Y. Levin, *J. Chem. Phys.* **142**, 194104 (2015).
- ⁶⁵F. A. Plamper, A. Walther, A. H. E. Muller, and M. Ballauff, *Nano Lett.* **7**, 167 (2007).
- ⁶⁶S. Alexander, P. M. Chaikin, P. Grant, G. J. Morales, and P. Pincus, *J. Chem. Phys.* **80**, 5776 (1984).
- ⁶⁷E. Trizac, L. Bocquet, M. Aubouy, and H. von Grunberg, *Langmuir* **19**, 4027 (2003).
- ⁶⁸B. Das, X. Guo, and M. Ballauff, *Prog. Colloid Polym. Sci.* **121**, 34 (2002).